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特殊材料之第一原理多體計算(2/3)

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行政院國家科學委員會專題研究計畫成果報告

特殊材料之第一原理多體計算(2/3)

Ab initio many-body calculations on novel materials (2/3)

計畫編號：NSC 93-2112-M-032 -004

執行期限：94年08月01日至95年07月31日

主持人：薛宏中 淡江大學物理系 副教授

一、中文摘要

在本計畫之第二階段，首先我們通過第一原理準粒子(quasiparticle)計算法，在GW近似之下，得到InN之能隙與最近實驗所得相當吻合；另一方面，由自旋極化密度泛函理論計算(spin-polarized DFT)，發現不僅在III-V族、II-VI族半導體或各種螺旋性質(chirality)之氮化硼奈米管(BNT)中，晶格空缺(vacancy)皆可能導致系統之磁偶極增大，進而造成磁性半導體。此外，我們也分析鈣鈦礦摻雜系統($Pb_{1-x}Ca_xTiO_3$)之電子結構與聲子振動模式，以釐清其相變機制。最後，我們將進行光譜之多體效應GW-BSE測試計算。

關鍵詞：準粒子 GW 近似、自旋極化密函、氮化硼奈米管、鈣鈦礦、GW-BSE

Abstract

At the 2nd step of this 3-year project, we perform quasiparticle calculations of InN within a GW approximation. Our spin-polarized DFT calculated electronic structures are in a good agreement with recent observation. On the other hand, our calculations indicate the possibility of vacancy-induced magnetization in not only III-V and II-VI compounds but also BN nanotubes in various chirality symmetries. Furthermore, we calculated the electronic structure and vibrational behavior of a perovskite with doping ($Pb_{1-x}Ca_xTiO_3$) in order to elucidate the mechanism of the doping effects on the structural phase transition. Finally, we start to test the optical spectrum calculation in GW-BSE scheme.

Keywords: quasiparticle, spin-polarized, BN nanotube, perovskite, GW-BSE

I、Introduction

Until recent measurement on MBE-grown wurtzite InN[1], the fundamental gap decreases to about 0.8 eV rather than a wider bandgap of 1.9 eV which has been accepted in last 20 years. In order to resolve this debate, we present *ab initio* quasiparticle band structure calculations for InN within GWA. The detail of this many-body perturbation theory can be found elsewhere [2].

On the other hand, just similar to the defect-induced magnetization in BN and AlN in our 1st year result, we extend this calculation systematically to III-V, II-V semiconductors and BN nanotubes (BNT) with versatile structures. Using spin-polarized DFT calculation taking into account the local geometrical distortion around the vacancy center, we will compare the magnetization corresponding to different compounds.

Furthermore, recent Raman and x-ray data showed a tetragonal→cubic→orthorhombic sequence of phase transition along with the concentration of doping in Ca-modified $PbTiO_3$ [3]. We performed DFT and DFPT calculation to study the structural stability of $Pb_{1-x}Ca_xTiO_3$.

Finally, in order to carry out the optical spectrum including the exciton effects, we will start to perform the GW-BSE calculation in which a 2-particle Green's function is adopted to represent the quasiparticles of electron-hole pairs.

II · Result and discussion

As shown in Fig. 1, comparing with negative direct bandgap(-0.02eV) at Γ point obtained from LDA, our GWA opens up the LDA bandgap up to 0.9 eV which is in good accord with recent measurements (~0.8 eV).

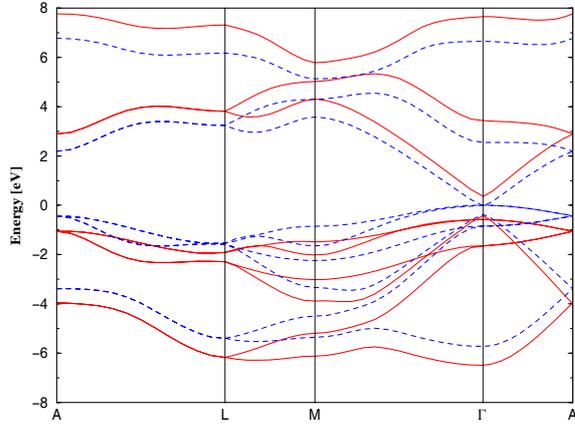


Fig.1. *Electronic band-structure of wurtzite-InN in (a) LDA (blue dashed curves) and (b) GWA (red solid curves).*

Our spin-polarized calculations also show that a vacancy of Ga and N in cubic GaN induces a nonzero magnetic moment of $3 \mu_B$ and $1 \mu_B$, respectively (as shown in Fig. 2(a) and (b)). Similar vacancy effect can be found in other III-V and II-VI compounds (as shown in Fig. 2 (c) and Table 1)

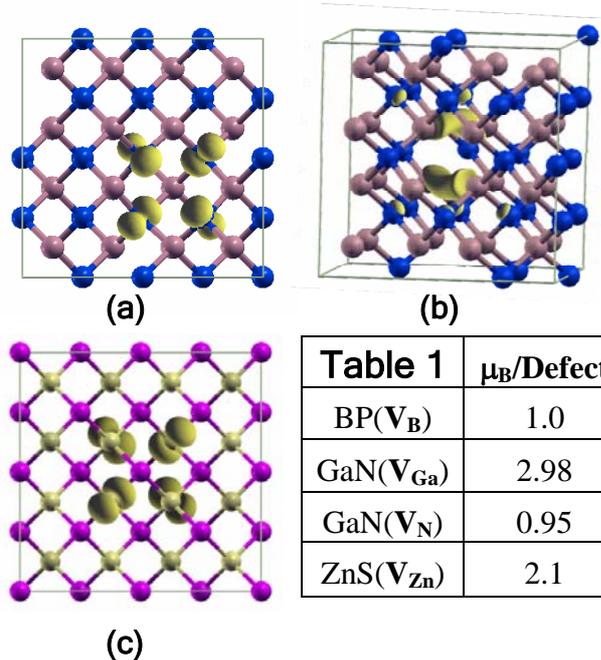


Fig. 2 *Electronic spin density around defect center of (a) V_{Ga} in GaN (b) V_N in GaN, and (c) V_{Zn} in ZnS. Corresponding magnetic moments are compiled in Table 1.*

Furthermore, we calculate the zigzag ((5,0)) armchair((5,5)), and chiral ((5,2)) BN nanotubes with the vacancy of B. As shown in Fig.3., the optimal structures near vacancy center all shows an outward expansion. The vacancy-induced magnetizations are all about $3 \mu_B/\text{vacancy}$.

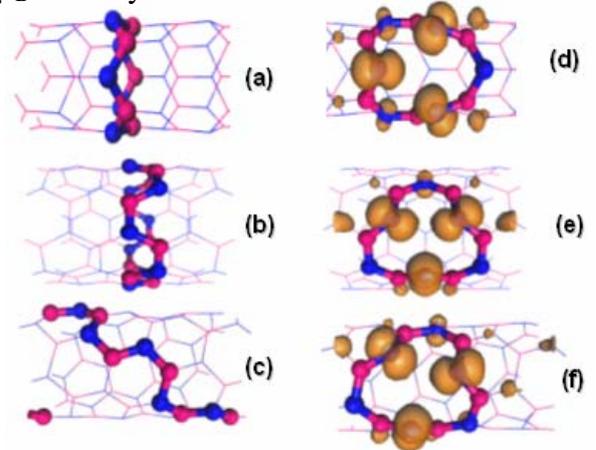


Fig.3. *Atomic geometry of BN nanotube in (a) (5,0) (b) (5,5) (c) (5,2) chiral symmetry. Corresponding optimal structure and spin density with a vacancy of B is shown in (d), (e), and (f), respectively.*

On the other hand, the electronic and vibrational structures of $Pb_{1-x}Ca_xTiO_3$ have been calculated to study the doping effects. Different bonding characters corresponding to different stable structures in $x=0.25$ and $x=0.5$ are shown in Fig. 4 (a) and (b).

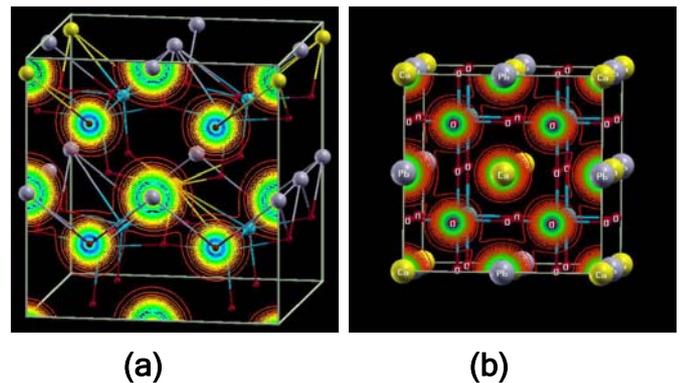


Fig. 4. Contour of charge density of (a) tetragonal $Pb_{0.75}Ca_{0.25}TiO_3$ and (b) cubic $Pb_{0.5}Ca_{0.5}TiO_3$. Pb, Ca, Ti, and O atom is denoted by a grey, yellow, light blue, and red sphere, respectively.

Based on the Density Functional Perturbation Theory (DFPT), we performed the phonon calculation for $Pb_{0.5}Ca_{0.5}TiO_3$. The eigenvector w.r.t. the softest mode at Γ point which dominates the structural instability is shown in Fig. 5. Such vibrational mode indicates the possibility of a stable orthorhombic structure for $x > 0.5$.

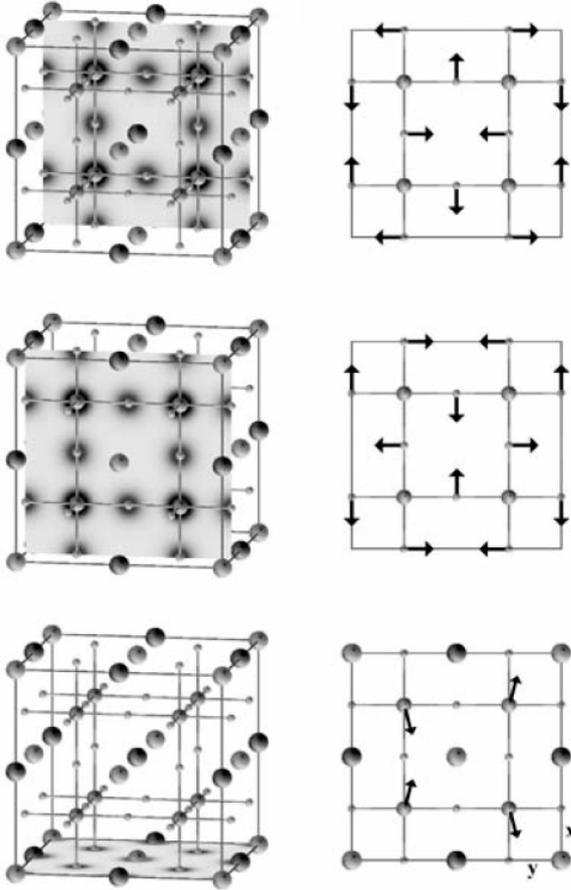


Fig.5. Eigenvectors, shown as arrows in arbitrary unit, of the softest mode at Γ point of $Pb_{0.5}Ca_{0.5}TiO_3$.

III Further work

In order to calculate the optical spectrum

with taking into account the electron-hole interaction, we start quasiparticle excitation calculation based on a scheme of 2-particle Green's function of Bathe-Salpeter equation [4]. Our preliminary results of Si, as shown in Fig. 6, represent the effect of electron-hole interaction (exciton) on optical spectrum which does not included in both DFT and the random phase approximation (RPA)-GWA calculation. We will carry out such GW-BSE calculations in other novel materials in our next year project.

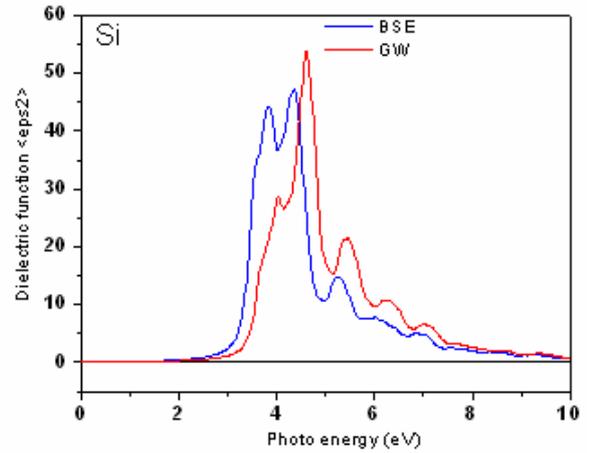


Fig.6 The calculated optical spectrum (ϵ_2) in RPA-GW (red curves) and GW-BSE (blue curves) scheme.

IV. References

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